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# **Conjugated ionic (co)polythiophene-based cathode interlayers for bulk heterojunction organic solar cells**

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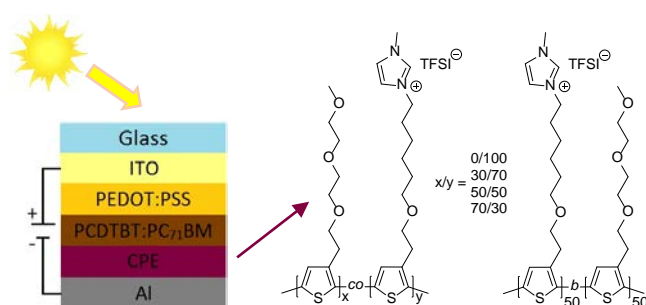
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## Abstract

The incorporation of conjugated polyelectrolytes as cathode interlayers in organic photovoltaics has been proven to be an effective way to boost the device efficiency. Nevertheless, more detailed investigations of the structure-property relationships of these interlayer materials, in particular related to the film deposition behavior, can provide further insights into their mode of action. With this aim, a series of ionic (co)polythiophenes is successfully synthesized via Kumada catalyst-transfer condensation polymerization and subsequent introduction of ionic moieties on the polymer side chains. Both the topology (i.e. homopolymers, random and block copolymers) and the amount of ionic groups are systematically varied. The polymers are fully characterized and then applied as cathode interlayers in polymer solar cells based on PCDTBT:PC<sub>71</sub>BM, affording an average efficiency increase of ~15%. The structural screening on one hand indicates that the efficiency gain is a rather general phenomenon for this material class. On the other hand, the best photovoltaic responses are observed for the conjugated polyelectrolytes with a higher triethylene glycol side chain ratio and the block copolymer structure performs slightly better as compared to the random copolymer with the same (50/50) monomer ratio. Based on these findings, the field can move on to a more rational development of novel interfacial materials and thereby push the device efficiency even further.

**Keywords:** conjugated polyelectrolytes; polymer solar cells; interfacial materials; topology; structure-property relationships



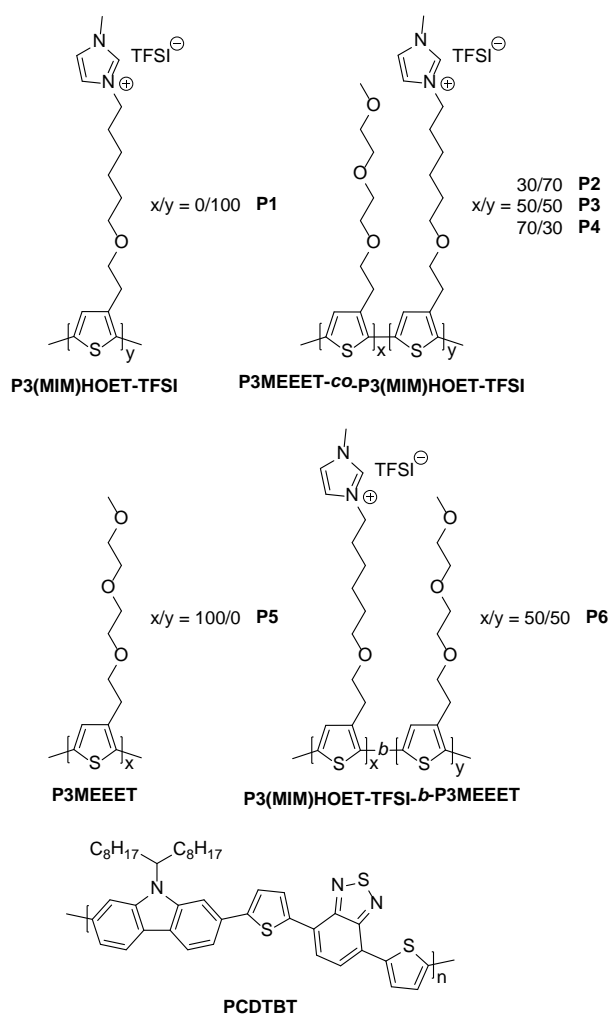
## Introduction

Over the past decade, organic photovoltaics (OPVs) have shown a growing potential as an alternative renewable energy source, in particular because of some additional appealing features such as flexibility, semi-transparency, improved low-light performance, and low-cost (high-throughput) large area production.<sup>1-3</sup> The introduction of the bulk heterojunction (BHJ) concept, in which the photoactive layer consists of electron donating and accepting materials intimately mixed at the nanoscale, the development of push-pull type low bandgap organic semiconductors to enhance the solar spectrum coverage, and the use of optimized device architectures have been key developments to elevate the power conversion efficiency (PCE) of OPV devices to values currently exceeding 12%.<sup>4-8</sup> Even though most research activities have been oriented on the development of novel active layer materials with appropriate optoelectronic properties, the introduction of interlayer materials has become a widely accepted approach to further enhance the device efficiency.<sup>9-17</sup> One class of interlayer materials of particular interest are conjugated polyelectrolytes (CPEs). They can be processed from eco-friendly, orthogonal solvents, thereby preventing re-dissolution of the underlying layer during device fabrication. The incorporation of such CPEs has led to PCE improvements of more than 20% compared to reference devices, which can mainly be attributed to the formation of interfacial dipoles, influencing the work function of the adjacent electrode.<sup>10,18-23</sup> This improves the energy level alignment between the photoactive layer and the cathode (in standard device stacks), resulting in an enhanced charge collection and a reduced charge recombination.

Several CPE materials have already been studied as cathode interlayers. The conjugated polymer backbone has not been explored much further than polyfluorene and polythiophene derivatives and most variation has been introduced in the ionic moieties.<sup>12</sup> Previous work in our group focused on imidazolium-functionalized polythiophenes, outperforming the

ammonium-functionalized counterparts.<sup>11</sup> More recently, impedance spectroscopy measurements revealed that the dielectric permittivity, induced by the ionic functionalities, is an important parameter to improve charge collection.<sup>23</sup> The exact CPE structure also influences other aspects. An ionic random (50/50) copolythiophene with triethylene glycol and imidazolium-functionalized side chains (**P3**; Figure 1) for instance demonstrated superior performance in comparison to a fully imidazolium-functionalized homopolymer, resulting from an improved affinity with the photoactive layer beneath due to the presence of the non-ionic side chains.<sup>23</sup> For inverted OPV devices incorporating the CPE at the bottom side, Chen *et al.* demonstrated that more smooth and homogeneous photoactive layer deposition could be obtained by increasing the amount of polar groups in polyfluorene-based CPEs.<sup>22</sup>

In this work, a series of polythiophene CPEs with TFSI (bis(trifluoromethane)sulfonimide) counteranions was synthesized by Kumada catalyst-transfer condensation polymerization (KCTCP), subsequent substitution of the bromide-functionalized side chains, and final exchange of the Br counteranions. Due to the controlled chain-growth character of this type of polymerization, different polymer topologies (homopolymers, random and block copolymers) were obtained with a high control over the molar mass, dispersity and end groups in a relatively straightforward way via one-pot procedures. The ratio of the ionic groups present in the random copolymers was systematically varied and a block copolymer was prepared as well for comparison. The synthesized materials were then evaluated as cathode interlayer materials in standard BHJ OPV devices based on PCDTBT:PC<sub>71</sub>BM to further explore the effect of structural changes on the overall device performance.



**Figure 1.** Overview of the synthesized polymers and the used donor material.

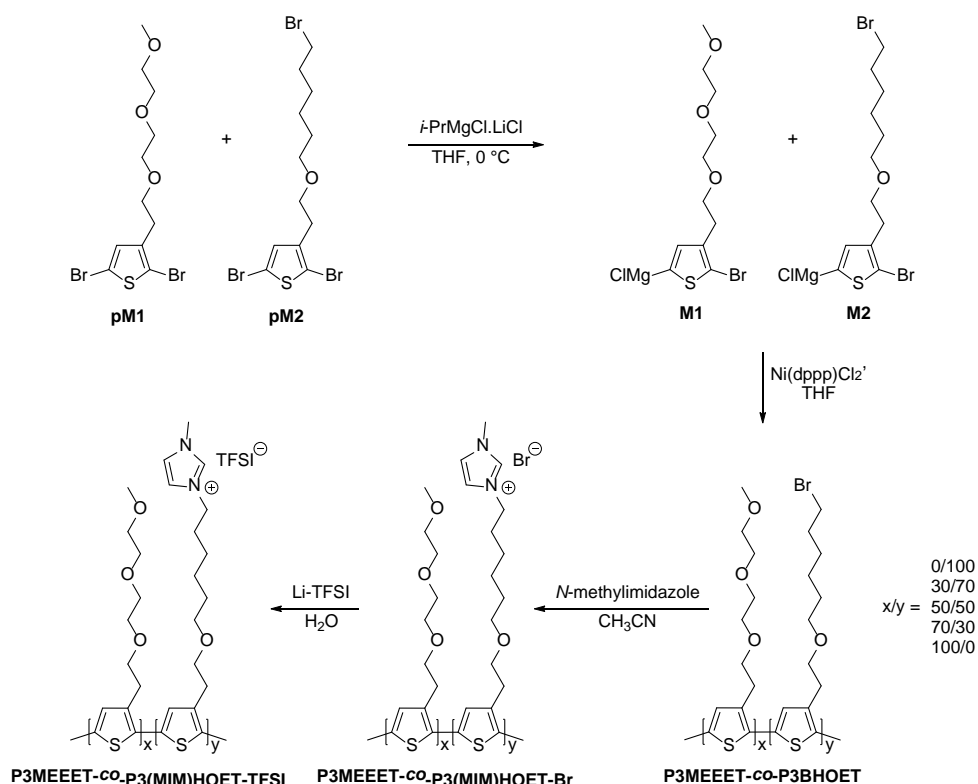
## Results and discussion

### Polymer synthesis

From previous work, the **P3MEEET-co-P3(MIM)HOET-TFSI** (50/50) random copolymer (**P3**; Figure 1) was identified as the CPE material affording best OPV device efficiencies for various photoactive layer blend combinations.<sup>23–25</sup> This material was hence used as the starting point to create further structural variation. A series of random copolymers with monomer ratios varying from 0/100, 30/70, 50/50, 70/30 to 100/0 was targeted. For the synthesis of these random copolymers, the two different precursor monomers were first prepared, i.e. 2,5-dibromo-3-[2-(2-(2-methoxyethoxy)ethoxy)ethyl]thiophene (**pM1**) and 2,5-

dibromo-3-[2-(6-bromohexyloxy)ethyl]thiophene (**pM2**) (Scheme 1).<sup>26</sup> These precursor monomers were then activated via a Grignard metathesis (GRIM) reaction. This was done prior to the polymerization by combining **pM1** and **pM2** in the respective ratio in dry THF, whereby a monomer concentration of 0.1 M was maintained, and adding 1.0 equivalents of *i*-PrMgCl.LiCl to this mixture at 0 °C. After stirring for 1 h, the active monomers **M1** and **M2** were obtained and transferred to a flask with the Ni(dppp)Cl<sub>2</sub> catalyst in dry THF (monomer concentration = 0.075 M) to start the polymerization (Scheme 1). Due to the controlled character of the KCTCP, the amount of Ni-catalyst could be adjusted for each individual polymerization to always end up with a number-average molar mass (*M<sub>n</sub>*) of ~25 000 g/mol.

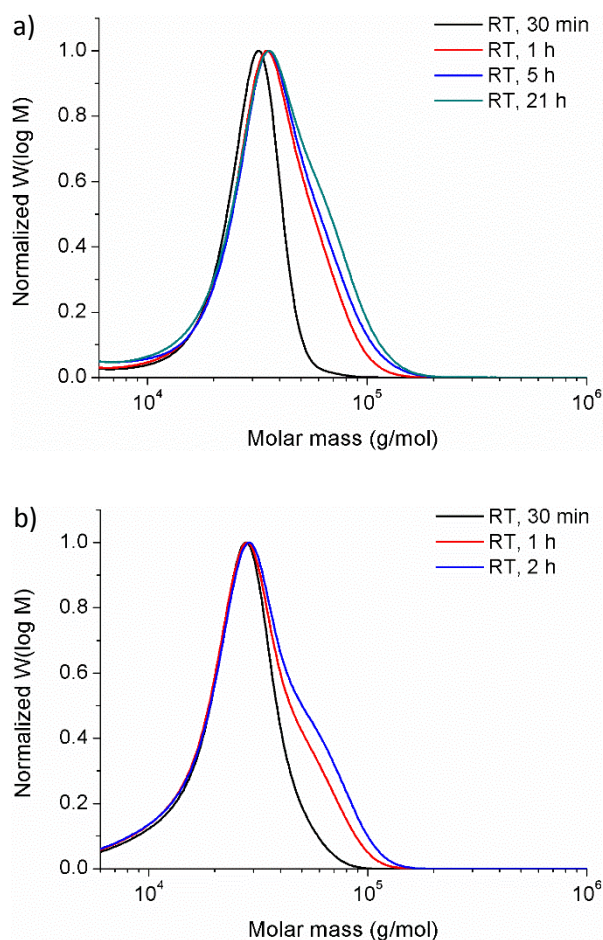
**Scheme 1.** Synthesis of the ionic (co)polythiophenes via KCTCP and two post-polymerization reactions.



To optimize the polymerization conditions and to be able to use a uniform polymerization method for the synthesis of all ionic (co)polythiophenes, the polymerization conditions for the two homopolymers, **P3BHOET** (poly[3-(2-(6-bromohexyloxy)ethyl)thiophene-2,5-diyl]; x/y

= 0/100) and **P3MEEET** (poly[3-(((methoxyethoxy)ethoxy)ethyl)thiophene-2,5-diyl]; x/y = 100/0) were investigated first. In a first polymerization test, **M2** was polymerized at RT and aliquots of the polymerization mixture were taken after 30 minutes, 1 hour, 5 hours, and 21 hours, and quenched with a MeOH/HCl mixture. These aliquots were then analyzed by gel permeation chromatography (GPC) and the results are shown in Figure 2a. From the GPC profiles it can be seen that a shoulder arises after 30 minutes, with a molar mass that is about two times the molar mass of the main peak, resulting in a rise in dispersity ( $\mathcal{D}$ ) from 1.14 to 1.34. The appearance of this shoulder can be explained by the occurrence of disproportionation, whereby two polymer chains are coupled to each other.<sup>27</sup> To evaluate whether this is also the case for the polymerization of **P3MEEET**, **M1** was also polymerized at RT and aliquots were taken after 30 minutes, 1 hour and 2 hours. The resulting GPC profiles (Figure 2b) once again show a shoulder arising after 30 minutes, leading to a rise in dispersity from 1.31 to 1.50.





**Figure 2.** GPC profiles of the polymerization tests at RT of a) **M2** and b) **M1**.

The above results indicated that a polymerization time of 30 minutes is optimal to obtain the desired molar mass and to avoid undesired termination reactions. Due to these findings, all polymerizations described in Scheme 1 were performed at RT and stopped after 30 minutes by quenching with a MeOH/HCl mixture. Afterwards, the (co)polymers were purified by Soxhlet extractions with MeOH, acetone (for **P3BHOET** (0/100), **P3MEEET-co-P3BHOET** (30/70) and **P3MEEET-co-P3BHOET** (50/50)), hexanes, and chloroform, respectively. For the polymers with a higher **P3MEEET** content, it was not possible to use acetone for the Soxhlet extraction since these polymers readily dissolve in acetone. After final precipitation in MeOH, the **P3MEEET-co-P3BHOET** random copolymers were obtained as purple solids. The resulting molar masses and dispersities are listed in Table 1. The built-in ratios of the

respective monomers were confirmed in the  $^1\text{H}$  NMR spectra of the polymers (Figures S3–S6, Supporting Information).

**Table 1.** GPC data of the synthesized polymers.

Polymer (x/y)	$M_n \times 10^4$ (g/mol)	$D$
<b>P3BHOET</b> (0/100)	1.7	1.18
<b>P3MEEET-co-P3BHOET</b> (30/70)	2.4	1.23
<b>P3MEEET-co-P3BHOET</b> (50/50)	2.8	1.16
<b>P3BHOET-b-P3MEEET</b> (50/50)	2.9	1.34
<b>P3MEEET-co-P3BHOET</b> (70/30)	2.6	1.17
<b>P3MEEET</b> (100/0)	1.9	1.30

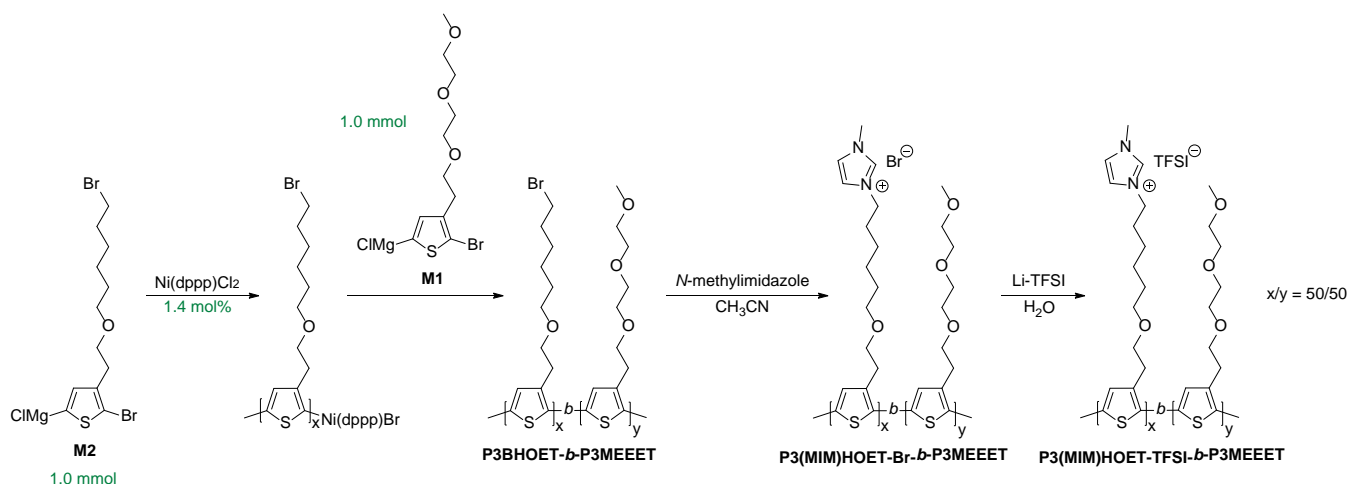
After obtaining the random copolymers, two post-polymerization reactions were performed to obtain the desired ionic (co)polythiophenes (Scheme 1). In the first reaction, the bromide groups at the end of the alkyl chains were replaced by *N*-methylimidazolium moieties. This was done *via* reaction with a large excess of *N*-methylimidazole in acetonitrile under microwave irradiation for 4 hours at 100 °C. The resulting ionic **P3MEEET-co-P3(MIM)HOET-Br** (co)polymers were then precipitated in Et<sub>2</sub>O and purified by Soxhlet extractions in Et<sub>2</sub>O and MeOH (dissolving the polymers), respectively, to remove the excess of *N*-methylimidazole. This large excess of *N*-methylimidazole was used to ensure that all bromide entities were converted to *N*-methylimidazolium moieties, which was confirmed by  $^1\text{H}$  NMR spectroscopy (Figures S7–S11, Supporting Information).

It was already shown before that ionic (co)polymers with bromide counteranions are strongly hygroscopic and as such not ideal for device incorporation.<sup>23,26,28</sup> Therefore, the bromide counteranions were replaced by more hydrophobic TFSI counteranions in a final step. This was performed by dissolving the **P3MEEET-co-P3(MIM)HOET-Br** (co)polymers in water, followed by a dropwise addition of a Li-TFSI solution in water (Scheme 1). After a few hours, the resulting precipitates were filtered off over a cellulose membrane, washed with water and dried under vacuum. In this way, the desired ionic **P3MEEET-co-**

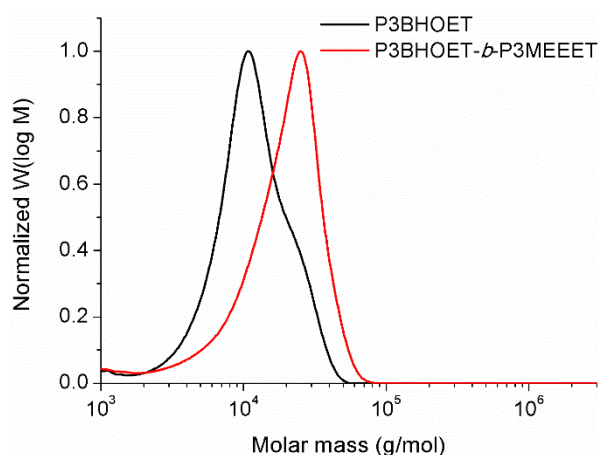
**P3(MIM)HOET-TFSI** (co)polymers were finally obtained. Full counteranion exchange was confirmed by  $^1\text{H}$  NMR spectroscopy.

Since the solar cell performance among others depends on the morphology of the interlayer material on top of the photoactive layer and block copolymers are known for their peculiar self-assembly properties,<sup>28–30</sup> a block copolythiophene was synthesized as well from the same two monomer units in a 50/50 ratio (Scheme 2). This was done by first polymerizing the **P3BHOET** block by adding one equivalent of *i*-PrMgCl.LiCl to **pM2** at 0 °C, resulting in the active monomer **M2** after 1 h, and then adding this reaction mixture to 1.4 mol% of Ni(dppp)Cl<sub>2</sub> at RT. After a polymerization time of 15 minutes, a small fraction of the polymerization mixture was quenched with a MeOH/HCl mixture, affording the **P3BHOET** reference homopolymer. To the rest of the polymerization mixture, the *in situ* prepared active monomer **M1** was added to create the **P3MEEET** second block. The polymerization of this second block was allowed to further proceed for 30 minutes at RT, before being quenched with a MeOH/HCl mixture. The resulting block copolymer was then purified by Soxhlet extractions with MeOH, hexanes, and chloroform, respectively, finally yielding the **P3BHOET-*b*-P3MEEET** block copolymer.

**Scheme 2.** Synthesis of the ionic **P3(MIM)HOET-TFSI-*b*-P3MEEET** block copolymer.



To verify the block copolymer formation, both the **P3BHOET** homopolymer sample and the **P3BHOET-*b*-P3MEEET** block copolymer were analyzed by GPC. From the resulting GPC profiles, it can be observed that there is a clear shift to higher  $M_n$  values for the block copolymer in comparison to the homopolymer (Figure 3). The **P3BHOET** homopolymer showed an  $M_n$  of 10 000 g/mol and a  $D$  of 1.38, while the **P3BHOET-*b*-P3MEEET** block copolymer has an  $M_n$  of 15 000 g/mol and a  $D$  of 1.42 (before Soxhlet extraction). After Soxhlet extraction, the block copolymer showed an  $M_n$  of 29 000 g/mol and a  $D$  of 1.34 (Table 1), nicely corresponding to the expectations and specifically targeted to be in range with the random copolymers described above. By using 1.4 mol% of  $\text{Ni(dppp)Cl}_2$ , an  $M_n$  of ~26 000 g/mol (~50 monomer units of 289 g/mol for the **P3BHOET** block and ~50 monomer units of 228 g/mol for the **P3MEEET** block) was foreseen for the **P3BHOET-*b*-P3MEEET** block copolymer (taking into account a monomer loss of ~25% due to the non-selective GRIM reaction).

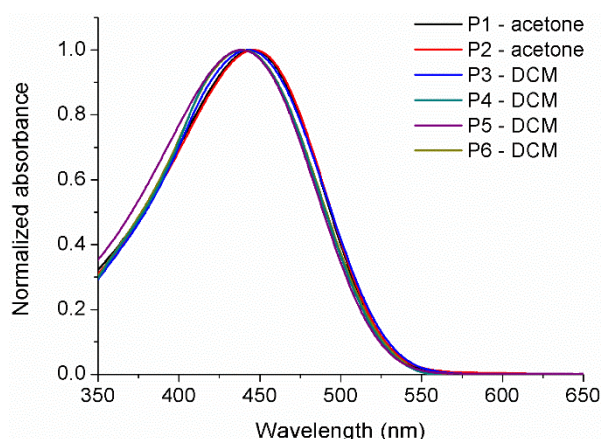


**Figure 3.** GPC profiles for the **P3BHOET** homopolymer (quench) and the **P3BHOET-*b*-P3MEEET** block copolymer (before Soxhlet extraction).

Finally, the obtained block copolymer was made ionic, following a similar procedure as described above, resulting into the ionic **P3(MIM)HOET-TFSI-*b*-P3MEEET** block copolymer (Scheme 2). An overview of the final synthesized ionic copolythiophenes is given in Figure 1. For the sake of clarity, the polymers are denoted as **P1–P6**, which will be used further on.

### Polymer characterization

The optical properties of the synthesized polymers were investigated by UV-vis absorption spectroscopy. Because of the different amounts of incorporated ionic groups, the polymers showed varying solubility characteristics and different solvents had to be used. From the spectra in Figure 4, it is clear that all polymers show about the same absorption profile, with a wavelength at maximal absorbance ( $\lambda_{\text{max}}$ ) located between 438 and 446 nm.

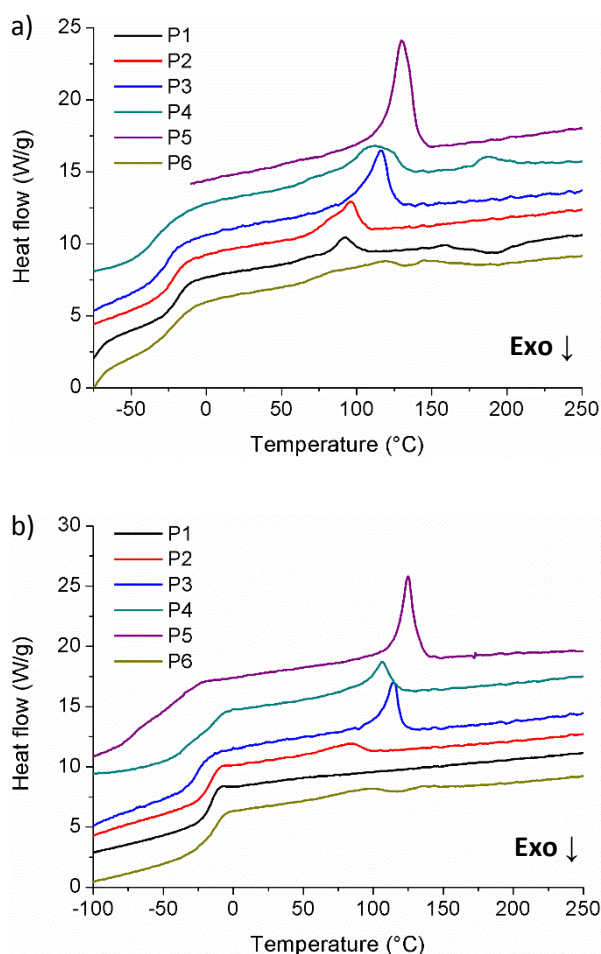


**Figure 4.** UV-vis absorption spectra in solution for the synthesized ionic polythiophenes **P1–P6**.

The electrochemical properties of the novel materials were analyzed by cyclic voltammetry (Table S1 and Figure S17, Supporting Information). P3HT-like behavior was observed for all materials. Detailed comparison of the obtained oxidation and reduction onsets and resulting HOMO-LUMO values is, however, complicated by the different solubility characteristics, impeding analysis of the full polymer series in one single solvent or in film.

The thermal stability of all polymers was studied by thermogravimetric analysis (TGA), showing strong similarities for all polymers and no significant mass loss up to 325–350 °C (Figure S18, Supporting Information). Further detailed thermal analysis was performed by rapid heat-cool calorimetry (RHC) measurements (Figure 5; Table S2–S3 and Figure S19, Supporting Information). From these data, it is clear that the **P1** ionic homopolymer shows a relatively low glass transition temperature ( $T_g$ ) and a low melting enthalpy ( $\Delta H$ ) after the first heating cycle. This melting behavior is, however, no longer observed when the polymer is reheated (second heating at 500 K/min) after a preceding cooling step at a relatively low cooling rate (20 K/min), implying that this cooling rate is too high to allow the development of a crystalline phase. On the other hand, the fully TEGylated (TEG = triethylene glycol) **P5** homopolymer shows a clear melting behavior after the first as well as the second heating cycle. The crystallinity of the random copolymers **P2–P4** and the block copolymer **P6**

depends on the relative content of the two constituent monomers. By decreasing the content of **P3(MIM)HOET-TFSI**, which clearly has a lower  $\Delta H$  than **P3MEEET**, the melting enthalpy (and therefore the crystallinity) of the random copolymers increases. This is confirmed when the  $\Delta H$  values of **P2–P4** and **P6** are estimated based upon the relative content of the two monomers and the  $\Delta H$  values of the homopolymers **P1** and **P5** after the first heating cycle (Table S2, Supporting Information). The melting temperatures ( $T_m$ ) and  $\Delta H$  values after the second heating cycle are consistently lower than after the first heating cycle (Tables S2 and S3, Supporting Information) which could be explained by the continued development of stable crystals during casting or at room temperature. Another important observation is that the 50/50 random copolymer **P3** and the 50/50 block copolymer **P6** show very different heating profiles. The block copolymer shows a very broad melting behavior with two different peaks, while the random copolymer shows just one distinct melting peak located in between those of the two homopolymers **P1** and **P5** (Figure 5a). This difference further confirms the formation of a block copolymer.



**Figure 5.** RHC heating profiles of the synthesized polymers **P1–P6**, obtained after a) a first heating cycle at 500 K/min, and b) a second heating cycle at 500 K/min (after preceding cooling at 20 K/min). The curves are shifted vertically for clarity.

## Photovoltaic properties

The incorporation of CPE interlayers in OPV devices has already been proven to be a successful strategy to boost the performance. In past work, we briefly demonstrated how the chemical structure of polythiophene-based CPE materials can influence the deposition behavior on top of different donor:acceptor blends.<sup>23</sup> To gain further insights into the film formation properties, the newly synthesized CPE materials were incorporated in BHJ polymer solar cells with a standard device architecture consisting of glass/ITO/PEDOT:PSS/PCDTBT:PC<sub>71</sub>BM/interlayer/Al (the PCDTBT structure is depicted



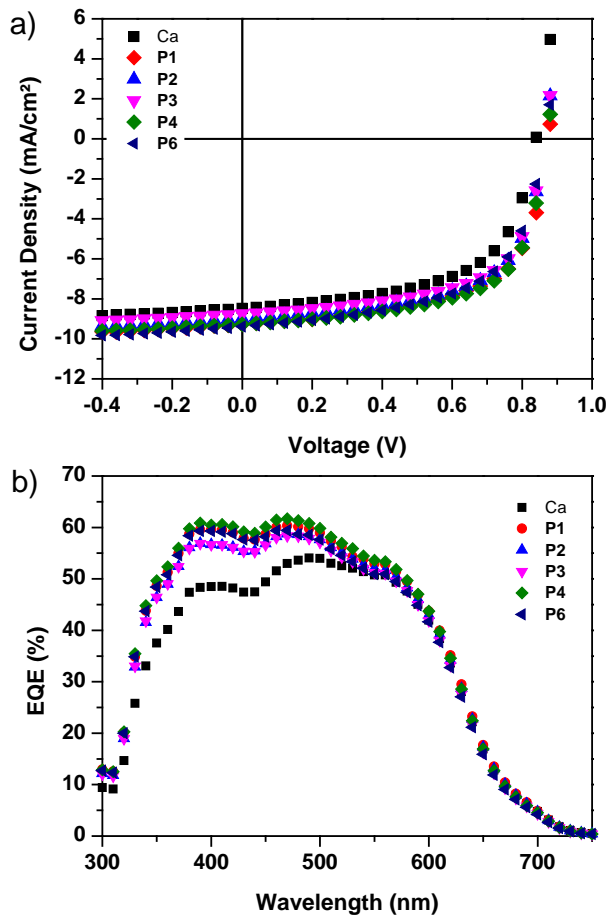
in Figure 1). The photoactive layer was prepared according to a literature recipe<sup>31</sup> with a total concentration of 20 mg/mL in *ortho*-dichlorobenzene (*o*DCB) and a 1:4 polymer:PC<sub>71</sub>BM ratio. The CPEs were deposited on top of the active layer from methanol solutions with varying concentrations. The device results are summarized in Table 2 and Figure 6a (the corresponding dark curves are depicted in Figure S20, Supporting Information). Polymer **P5** was excluded from the series because of its insoluble nature in the chosen solvent. As can be observed, the reference device employing calcium instead of a CPE exhibits an average PCE of 4.10%, with a top efficiency of 4.34%. Similar to previous studies on PCDTBT:PC<sub>71</sub>BM devices containing CPE interlayers,<sup>11,23</sup> all photovoltaic parameters (open-circuit voltage ( $V_{OC}$ ), short-circuit current density ( $J_{SC}$ ), fill factor (FF), and PCE) were enhanced, resulting in an average PCE increase of 15–17%. As such, the top performing device incorporating a CPE (**P4**) demonstrated, on average, a  $V_{OC}$  of 0.88 V (vs. 0.84 V), a  $J_{SC}$  of 9.02 mA/cm<sup>2</sup> (vs. 8.71 mA/cm<sup>2</sup>) and a FF of 62% (vs. 56%), resulting in an average PCE of 4.92% (vs. 4.10%). The increase in  $J_{SC}$  for all interlayer containing devices was also confirmed by extracting the currents from external quantum efficiency (EQE) measurements on average-performing devices (Figure 6b), with  $J_{EQE}$  values of 8.00, 8.81, 8.57, 8.44, 8.87, and 8.51 mA/cm<sup>2</sup> for the devices employing Ca, **P1**, **P2**, **P3**, **P4**, and **P6**, respectively (with corresponding  $J_{SC}$  values of 8.14, 8.79, 8.78, 8.77, 8.86, and 8.62 mA/cm<sup>2</sup>, respectively).

**Table 2.** Photovoltaic parameters of BHJ polymer solar cells (PCDTBT:PC<sub>71</sub>BM) with and without CPE cathode interlayers.

Entry	Cathode Interlayer	Concentration (mg/mL)	$V_{OC}$ (V)	$J_{SC}$ (mA/cm <sup>2</sup> )	FF	Average $\eta$ (%) <sup>a</sup>	Best $\eta$ (%)
1	Ca	/	0.84	8.71	0.56	4.10	4.34
2	<b>P1</b>	0.25	0.85	8.41	0.60	4.29	4.65
3	<b>P1</b>	0.5	0.87	8.90	0.58	4.49	4.93
4	<b>P1</b>	1	0.88	9.01	0.59	4.68	4.86
5	<b>P1</b>	1.5	0.86	9.09	0.61	4.77	5.11
6	<b>P2</b>	0.25	0.87	8.92	0.61	4.73	5.04
7	<b>P2</b>	0.5	0.88	8.86	0.61	4.76	4.76

8	<b>P2</b>	1	0.87	8.94	0.60	4.67	4.91
9	<b>P2</b>	1.5	0.87	9.04	0.61	4.80	4.98
10	<b>P3</b>	0.25	0.84	8.35	0.62	4.35	4.38
11	<b>P3</b>	0.5	0.86	8.76	0.61	4.60	4.83
12	<b>P3</b>	1	0.87	8.86	0.61	4.70	4.81
13	<b>P3</b>	1.5	0.87	8.56	0.61	4.54	4.96
14	<b>P4</b>	0.25	0.86	8.32	0.63	4.51	4.80
15	<b>P4</b>	0.5	0.85	8.35	0.61	4.33	4.64
16	<b>P4</b>	1	0.88	9.02	0.62	4.92	5.11
17	<b>P4</b>	1.5	0.88	8.98	0.61	4.82	5.05
18	<b>P6</b>	0.25	0.88	9.23	0.60	4.87	5.06
19	<b>P6</b>	0.5	0.88	9.21	0.60	4.86	5.14
20	<b>P6</b>	1	0.88	9.39	0.59	4.84	4.95

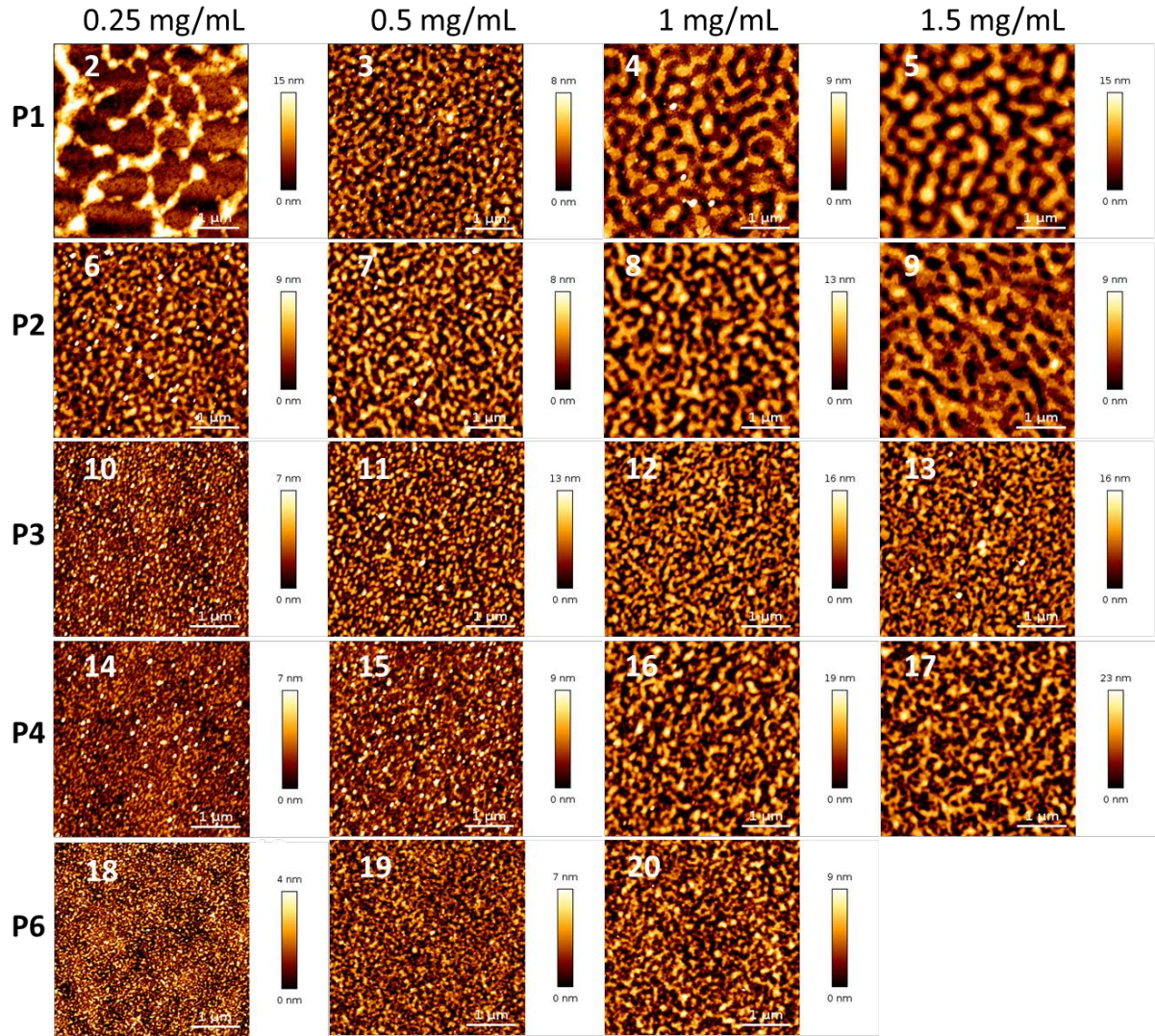
<sup>a</sup> Average PCE over 8–16 devices.



**Figure 6.** a) *J*-*V* curves under illumination for average performing PCDTBT:PC<sub>71</sub>BM BHJ

photovoltaic devices with and without CPE cathode interlayers. b) EQE spectra of average performing devices bearing Ca and CPE interlayers **P1–P6**.

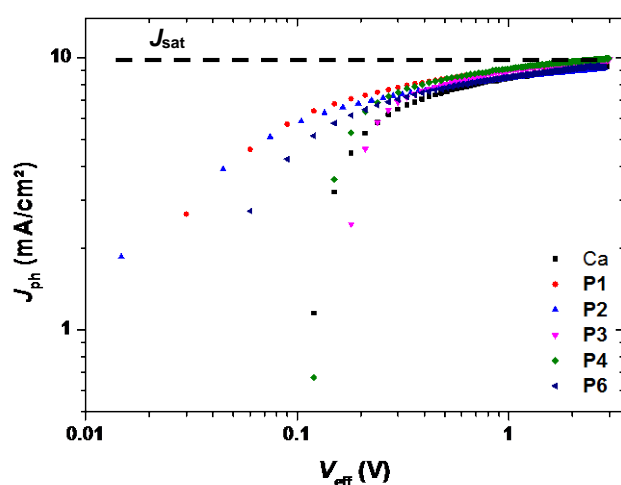
To investigate the interlayer deposition behavior in more detail, atomic force microscopy (AFM) imaging was performed on samples containing the CPE materials deposited on top of the PCDTBT:PC<sub>71</sub>BM active layer (Figure 7; the entry numbers from Table 2 were added for convenience). Similar to previous observations,<sup>11,23</sup> the polythiophene-based CPE interlayers never afford a completely covered layer, and the deposition patterns strongly vary depending on the CPE concentration (in methanol). Firstly, increasing the ratio of TEG-to-ionic side chains in the respective CPE materials results in an improved compatibility for deposition onto the photoactive layer blend, most profoundly observable for the samples with CPEs deposited from a concentration of 0.25 mg/mL. Moreover, especially for **P3** and **P4**, the lower device performances for the lower concentrations can be related to too narrowly distributed CPE domains (entries 10, 14 and 15), whereas slightly larger and interconnected domains appear to give rise to more elevated performances (entries 4, 5, 16 and 17). Furthermore, alteration of the chemical structure from the random copolymer **P3** to the block copolymer **P6**, with an identical TEG-to-ionic side chain ratio, also influences the photovoltaic parameters. Solar cells employing the **P6** interlayer demonstrated slightly superior performances across the entire concentration range in comparison to **P3**, mostly originating from an elevated  $J_{SC}$  (from an average of 8.65 to 9.28 mA/cm<sup>2</sup>), finally resulting into PCEs exceeding 5%. AFM imaging, however, did not reveal noteworthy differences between the deposition behavior of these two CPE interlayers.



**Figure 7.** Atomic force microscopy images (4x4  $\mu\text{m}$ ) of CPE materials **P1–P4** and **P6** deposited from various concentrations in methanol on top of the PCDTBT:PC<sub>71</sub>BM active layer. The image numbers refer to the entries in Table 2.

In previous work, we demonstrated that the polythiophene-based CPE interlayers form a capacitive double layer due to the presence of the ionic moieties, thereby enhancing the dielectric constant and charge carrier dissociation of the overall solar cell.<sup>23</sup> As a complementary technique, the charge dissociation probability ( $P$ ), determined by the correlation between the photocurrent  $J_{\text{ph}}$  and the effective voltage  $V_{\text{eff}}$ , can provide further insights into the exciton dissociation.<sup>32</sup> Herein,  $J_{\text{ph}}$  can be defined as  $J_{\text{L}} - J_{\text{D}}$ , in which  $J_{\text{L}}$  and  $J_{\text{D}}$  are the current densities under light and dark conditions, respectively, and  $V_{\text{eff}}$  can be

defined as  $V_0 - V$ , where  $V_0$  is the voltage at which  $J_{ph} = 0$  and  $V$  is the applied bias. As such, the saturation current  $J_{sat}$  was determined to be 9.26, 9.56, 9.15, 9.88, 9.93 and 9.29 mA/cm<sup>2</sup> for the reference device and the solar cells containing **P1**, **P2**, **P3**, **P4** and **P6**, respectively (Figure 8). From these values,  $P$  could be estimated by  $J_{ph}/J_{sat}$  to be 89.7, 94.4, 98.6, 91.1, 91.7 and 95.5%, respectively, under short-circuit conditions, confirming that the CPE bearing devices show an elevated exciton dissociation rate and an enhanced charge collection efficiency in comparison to the reference device.



**Figure 8.** Photocurrent of PCDTBT:PC<sub>71</sub>BM photovoltaic devices with and without CPE cathode interlayers as a function of the effective voltage.

## Conclusions

A series of **P3MEEET-co-P3(MIM)HOET-TFSI** random copolymers with different ratios of the ionic side chains (30/50/70%; **P2–P4**), the homopolymer counterparts **P1** and **P5**, and the 50/50 ionic block copolymer analogue **P6** were successfully synthesized via Kumada catalyst-transfer condensation polymerization and subsequent introduction of the ionic moieties and counteranion exchange. Because of the high degree of control that can be achieved in such a polymerization, all copolymers were obtained in the desired monomer ratios and with the targeted molar masses of ~25 000 g/mol. All materials demonstrated

similar absorption characteristics in solution. On the other hand, the polymer crystallinity clearly increased upon decreasing the amount of ionic groups.

The ionic (co)polythiophenes were then applied as cathode interlayer materials in BHJ photovoltaic devices based on PCDTBT:PC<sub>71</sub>BM. All photovoltaic parameters ( $V_{OC}$ ,  $J_{SC}$ , FF) improved throughout the complete series in comparison to a reference device employing Ca, pointing to the generality of the efficiency gain (on average 15–17%) for this interlayer class, regardless of the structural details. The elevated  $J_{SC}$  was confirmed by determining the charge dissociation probabilities of all devices. These measurements revealed an increase in  $P$ , and hence the exciton dissociation rate and charge collection efficiency, for all CPE bearing devices. AFM analysis of the final device stacks showed that the polythiophene-based CPE interlayers never afford a completely covered photoactive layer and that the deposition patterns strongly vary with the CPE concentration. The AFM images also illustrated that an improved compatibility for deposition onto the photoactive layer is obtained by increasing the ratio of triethylene glycol-to-ionic side chains. These results confirm our hypothesis that the presence of ionic groups induces the formation of interfacial dipoles, improving the charge collection, while at the same time reducing the affinity of the interlayer material for the photoactive layer, and that the increase in PCE is dependent on the interplay of these two phenomena.<sup>23</sup> An interesting observation is that the 50/50 block copolymer (slightly) outperforms the 50/50 random copolymer, which previously afforded record efficiencies.<sup>23–25</sup> Since it is known that control of the morphology is crucial for device performance and that block copolymers can self-assemble into nanostructured morphologies, it seems worthwhile to further explore this route.

## **Supporting information**

Instrumentation details, experimental (synthetic) procedures and characterization data as well as  $^1\text{H}$  NMR, CV, RHC, and TGA figures can be found in the Supporting Information.

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